

# Determination of the glass transition temperature of poly(styrene-co-divinylbenzene) by inverse gas chromatography

Roman Sanetra, Bożena N. Kolarz\* and Andrzej Wlochowicz

Textile Institute, Technical University of Łódź, Bielsko-Białá, Poland

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Inverse gas chromatography (i.g.c.) was used to characterize the supermolecular structure of modified, moderately crosslinked poly(styrene-co-divinylbenzene) (5–20 wt%) and to determine the interactions with normal alcohols ( $C_1$ – $C_6$ ). For all copolymers a decrease of retention time with increasing molecular weight of alcohols from  $C_1$  to  $C_3$  followed by an increase for  $C_4$  to  $C_6$  alcohols was observed. It was found that with propanol or butanol the measured  $T_g$  values were close to those measured by d.s.c. The  $T_g$  values for poly(styrene-co-divinylbenzene) containing 5, 10 and 20 wt% divinylbenzene were: 102°C, 116°C and 133°C respectively. The highest alcohols ( $C_5$  and  $C_6$ ) acted as plasticizers and reduced the apparent  $T_g$  value. The relationship between the isosteric heat of adsorption and the number of carbon atoms in the alcohols was non-linear with a minimum corresponding to butanol.

(Keywords: poly(styrene-co-divinylbenzene); gas chromatography; glass transitions; heat of adsorption)

## INTRODUCTION

The glass transition temperatures ( $T_g$ ) of many polymers have been measured using inverse gas chromatography (i.g.c.)<sup>1,2</sup>. This method exploits the temperature dependence of interactions between polymer and adsorbate, termed the molecular probe. At the glass transition point, the character of such a dependence changes, clearly affecting the measured retention volumes. Thus, the  $T_g$  is taken to be the first significant deviation of a point from the straight line determined by the retention volume,  $V_g$ , measured at  $T < T_g$  in the retention diagrams, i.e. in the plots  $\log V_g$  vs.  $1/T$ .

The polymer tested is usually deposited on glass beads or on adsorbents of Chromosorb, Poropak or other similar types of materials<sup>1–3</sup>. Measurements are also carried out with columns packed with a mixture of inert filling and polymer powder<sup>4</sup>. It has been shown that  $T_g$  does not depend on the physical form of the polymer<sup>2,4,5</sup>, e.g. on the thickness of deposited layer<sup>1,2,6</sup>, except for extremely thin layers, below 280 nm<sup>7</sup>. The i.g.c. determined  $T_g$  may, however, depend on the conditions of measurements and on the kind of molecular probe applied. Controversy still exists as to whether thermodynamically good or bad solvents, with respect to the polymer, should be used as probes<sup>1–19</sup>.

Highly crosslinked styrene-divinylbenzene (S-DVB) copolymers (with 35 wt% or more DVB) are widely used as chromatographic packing<sup>20–22</sup>. They are more commonly known under the trademarks Chromosorb, Poropak, etc. For these copolymers, the heats of adsorption of various probes, e.g. alcohols, were measured by i.g.c.<sup>20</sup> The adsorption isotherms of *n*-decane on the gel-type S-DVB copolymer containing 8% DVB were also measured by i.g.c.<sup>23</sup>

Here, we report on the results of our attempts to apply i.g.c. to the characterization of the supermolecular structure of modified, moderately crosslinked S-DVB copolymers and to determine their interaction with solvents. Firstly, we present the measurements in which  $C_1$ – $C_6$  normal alcohols are used as molecular probes.

## EXPERIMENTAL

The copolymers were prepared by suspension copolymerization of styrene (Z.Ch. Oświęcim, Poland) and commercial divinylbenzene (Merck, FRG). The latter contained 38.4% *m*-DVB, 14.8% *p*-DVB, 33.0% *m*-ethylstyrene, and 11.5% *p*-ethylstyrene, by weight. The monomers were copolymerized in the presence of toluene (T)–*n*-heptane (H) mixtures used as inert diluents. The polymerization procedure has been described in detail elsewhere<sup>24–28</sup>. The copolymers in the form of regular beads of diameter 80–120  $\mu\text{m}$  (cf. Table 2) were extracted with hot benzene and dried at 80°C–150°C for 120 h. As before, the composition of each sample is coded in its symbol as in the following example. The symbol 'HT 19.0.5/5' indicates that the ratio heptane:toluene was 1:9, by volume, the volume fraction of monomers (S+DVB) in the system was 0.5, and the last digit (5) denotes the weight per cent of *p*- and *m*-DVB in the S-DVB mixture.

For the apparent density and swelling ratio measurements the samples were prepared in the following way (cf. ref. 25). The copolymers, swollen in toluene, were transferred into a small column and treated successively with acetone, methanol, butanol and hexanol, at room temperature. After each solvent had passed through the column, a sample of copolymer was withdrawn and the swelling ratio,  $W$ , in grams of solvent per gram of dry copolymer, determined by centrifugation. The sample was then dried at 25°C  $\pm$  0.5°C to constant weight and its

\* Institute of Organic and Polymer Technology, Technical University of Wrocław, Wybrzeże Wyspiańskiego 27, 50-370 Wrocław, Poland.

**Table 1** Properties of poly(styrene-co-divinylbenzene)

Samples	$X$	Density ( $\text{g cm}^{-3}$ )		Porosity (%)	Toluene		Methanol		Butanol		Hexanol	
		$d$	$d_0$		$W$	$v_2$	$W$	$v_2$	$W$	$v_2$	$W$	$v_2$
HT 19 0.5/5	24	1.03	1.03	0	1.80	0.32	0.14	0.85	0.17	0.82	0.14	0.85
HT 19 0.5/10	11	1.06	1.01	4.7	1.11	0.64	0.18	0.81	0.23	0.77	0.23	0.77
HT 19 0.5/20	7	1.07	1.01	5.3	0.79	0.89	0.35	0.68	0.42	0.64	0.42	0.68

$X$ --number of styrene units between network junctions  
 $d$ --true density  
 $d_0$ --apparent density for samples pretreated with methanol  
 $W$ --solvent regain g/g  
 $v_2$ --volume fraction of polymer in swollen copolymer gels

apparent and true density measured according to procedures described before<sup>25,27</sup>.

The alcohol probes were chromatographic standards supplied by Poly-Science (USA) and POCh (Poland). 0.5  $\mu\text{l}$  of methanol, ethanol, n-propanol, n-butanol, n-pentanol, or n-hexanol were injected at each run. The copolymers were packed into chromatographic columns of 1 m length and  $\frac{1}{8}$ " internal diameter, and conditioned for 24 h at 150°C in hydrogen.

A Varian (Aerograph series 2860) chromatograph with a thermal conductivity detector was used. The flow rate of the hydrogen carrier gas was  $19 \pm 0.03 \text{ cm}^3/\text{min}$ . The retention times were measured to within 0.1 s. Methane was used as an inactive agent.

The isosteric heat of adsorption,  $q_{st}$ , was calculated from the slope of the least-squares line in the adsorption region of the retention diagrams (below  $T_g$ ), according to the relation<sup>29</sup>:

$$q_{st} = R \frac{d(\log V_g)}{d(1/T)} + RT$$

where  $T$  is the average temperature in this region and  $R$  is the gas constant.

## RESULTS AND DISCUSSION

The S-DVB copolymers were obtained by suspension copolymerization of the monomers in the presence of inert diluents. The degree of crosslinking was not too high: 5, 10 and 20 wt% of DVB (Table 1). The bead diameter was ca. 0.1 mm. The thickness of the polymer layer was thus  $10^3$  times higher than that for the samples deposited on fillings<sup>1,11</sup>.

Stoichiometry predicts a small number of S units between network junctions (DVB units) in the copolymers, but due to the intrinsic inhomogeneity of S-DVB copolymers, the linear segments remain mobile at temperatures only slightly higher than that for polystyrene. The inhomogeneities of S-DVB copolymers originate at the beginning of the copolymerization process. At that stage, DVB units react relatively faster than do the S units and hence precursors of highly crosslinked microgels are formed<sup>30-32</sup>. The microgel precursors solvated by unreacted monomers (and diluents) are then entangled and linked together by essentially linear segments rich in S units. Those segments are probably responsible for the overall segmental mobility<sup>27</sup> that determines  $T_g$  values of S-DVB copolymers<sup>33</sup>. The inhomogeneity due to this mechanism of polymer formation leads to peculiarities in

the behaviour of S-DVB copolymers containing more than ca. 10 wt% DVB and obtained in the presence of inert diluents, namely, that porous copolymers undergoing a microsineresis are formed. The microsineresis manifests itself as differences in apparent densities (porosities) of the same copolymers pretreated with various solvents. The extent to which such differences are observed is dependent on the initial composition of the mixture used for the copolymer preparation<sup>24-28</sup>.

No microsineresis was observed for the copolymer HT 19 0.5/5 (Table 1). The copolymers containing 10 and 20 wt% DVB are gels when dried out from toluene, but become porous when pretreated with a bad solvent. The apparent densities and porosities of the samples pretreated with methanol are given in Table 1. Further evidence for the microsineresis of the copolymer gel is provided by the solvent regain measurements. An increase in swelling ratio,  $W$ , in alcohols with an increase in the degree of crosslinking (Table 1) can hardly be explained otherwise than by assuming that new pores are being formed in bad solvents (alcohols). A local collapse of gel portions, not followed by shrinkage of the whole bead, seems to be responsible for such behaviour. For toluene, the swelling ratio decreases with increasing degree of crosslinking of copolymers as it should also for polymer networks.

The data on the swelling ratios and the volume fractions of polymer in swollen copolymer gels ( $v_2$ ) calculated therefrom confirm that methanol is the worst solvent for S-DVB copolymers; the swelling ratios for higher alcohols are somewhat higher, in accordance with the decreasing differences in the solubility parameters of the alcohol-polystyrene pairs (Table 3). One should notice, however, that at least two effects, namely, the swelling ability and formation of new pores, are superimposed on each other and both influence the swelling ratio measurements.

In the preliminary i.g.c. measurements, the effect of flow rate on the retention volumes has been measured at

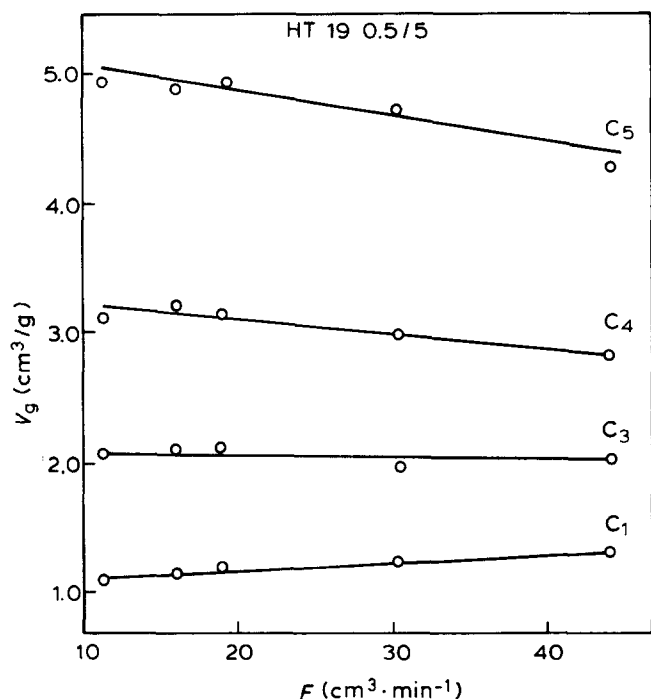
**Table 2** Column characteristics

Samples	Bead diameter (mm)	Weight of copolymer (g)	Surface of copolymer in i.g.c. column ( $\text{m}^2$ )
HT 19 0.5/5	$0.08 \pm 0.05$	4.2	0.030
HT 19 0.5/10	$0.12 \pm 0.06$	4.2	0.020
HT 19 0.5/20	$0.12 \pm 0.05$	4.8	0.022

**Table 3** Solubility parameters ( $\delta_1$ ) and the isosteric heats of adsorption ( $q_{st}$ ) for the alcohols

Alcohols	$\delta_1$ (MPa <sup>1/2</sup> )	$\delta_1 - \delta_2^a$ (MPa <sup>1/2</sup> )	$q_{st}$ (KJ mol <sup>-1</sup> )		
			5	10	20
C <sub>1</sub> CH <sub>3</sub> OH	29.7	11.1	13.3	14.3	11.7
C <sub>2</sub> C <sub>2</sub> H <sub>5</sub> OH	26.0	7.4	12.0	10.3	8.1
C <sub>3</sub> C <sub>3</sub> H <sub>7</sub> OH	24.3	5.7	9.1	7.9	6.5
C <sub>4</sub> n-C <sub>4</sub> H <sub>9</sub> OH	23.3	4.7	7.0	6.7	6.5
C <sub>5</sub> n-C <sub>5</sub> H <sub>11</sub> OH	22.3	3.7	9.4	7.2	6.4
C <sub>6</sub> n-C <sub>6</sub> H <sub>13</sub> OH	21.9	3.3	14.2	13.0	8.1

<sup>a</sup>  $\delta_2 = 18.6$  MPa<sup>1/2</sup> (ref. 35)



**Figure 1** Relationship between retention volume for normal alcohols at 182°C and rate of gas flow for HT 19 0.5/5 copolymer

182°C (mixed adsorption-absorption region). The results are shown in Figure 1. It can be seen that the flow rate in the range studied does not alter the results significantly.

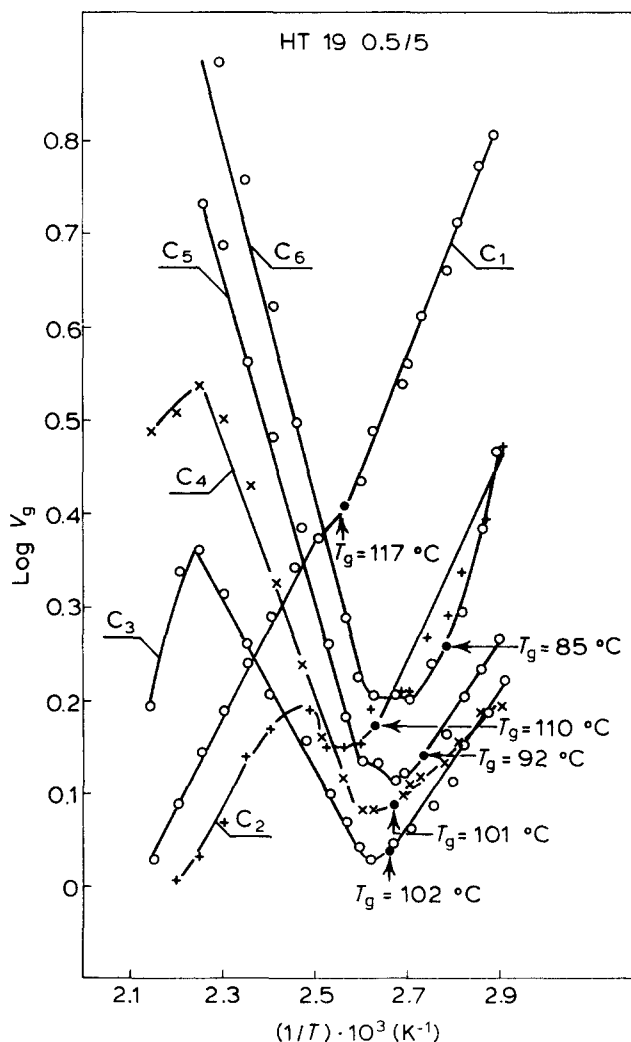
The retention diagrams for the HT 19 0.5/5 copolymer are presented in Figure 2. The retention volumes in the adsorption region (right-hand-side branches of curves) decrease from methanol to propanol, to increase again for higher alcohols.

The retention diagrams in Figure 2 are significantly different from those measured for nonspecific adsorbents such as graphitized carbon black or Poropak Q where the retention volumes increase smoothly with increasing molecular weight of alcohols<sup>20</sup>, although Poropak Q, being a copolymer of commercial DVB with high specific surface area (ca. 700 m<sup>2</sup>/g), is chemically similar to the copolymers being studied<sup>21</sup>. Probably the differences in supermolecular structures of the copolymers are responsible for their differing behaviour.

The log  $V_g$  vs.  $1/T$  curves in Figure 2 have different shapes for different alcohols. Methanol, which is considered to be a D-type adsorbate, weakly interacts with

the copolymers. In almost all ranges of temperatures its retention diagrams are linear, similar to those for the hexadecane-polystyrene system, with the latter being in the form of a thin layer deposited on an inert support<sup>1,17</sup>. For ethanol, the retention curves are similar to those measured for thick layers of polymer<sup>17</sup>, but already for propanol, the non-equilibrium adsorption-absorption range becomes very broad. In the range of temperatures studied, the absorption equilibrium for C<sub>5</sub> and C<sub>6</sub> alcohols was not reached. Because of small diffusion coefficients through the copolymer gel the probes are unable to penetrate through the whole volume of beads and an increase in retention volume is observed. The non-equilibrium region was found to be broader the higher was the degree of crosslinking of the copolymer. The effect of crosslinking degree on the retention diagrams for ethanol and butanol is presented in Figures 3 and 4.

The results of the regression analysis applied to the adsorption region (below  $T_g$ ) are presented in Table 4. As can be deduced from Figure 2, the slopes of the lines and therefore the values of the isosteric heats of adsorption,  $q_{st}$ , decrease for alcohols in the order C<sub>1</sub> > C<sub>2</sub> > C<sub>3</sub> > C<sub>4</sub> and then increase for the higher alcohols, following a roughly similar relationship for polymer-solvent interaction expressed by differences in the solubility parameters (cf. Figure 5).



**Figure 2** Retention diagram for normal alcohols as molecular probes with HT 19 0.5/5 copolymer

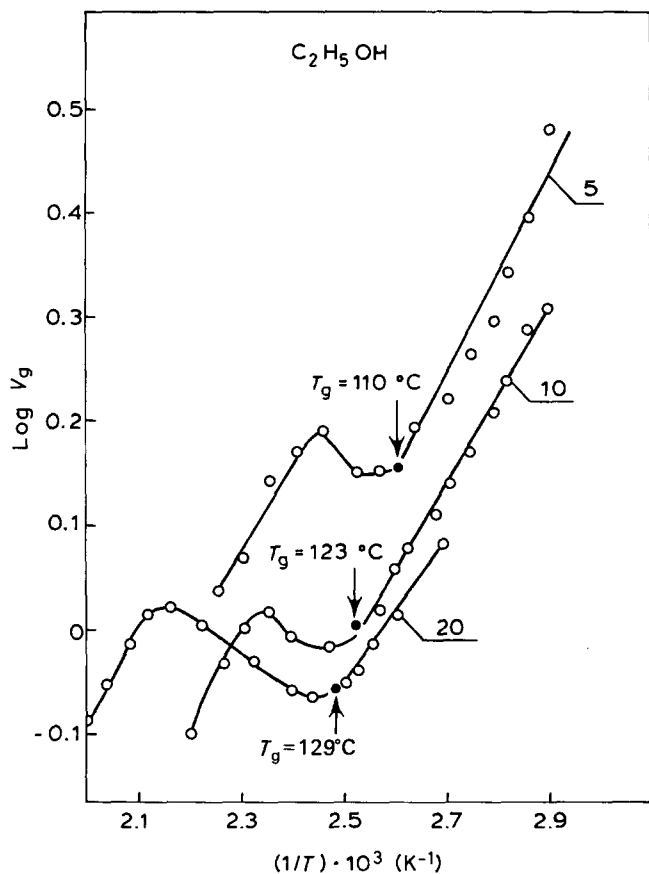


Figure 3 Retention diagram for S-DVB copolymers and ethanol as molecular probe

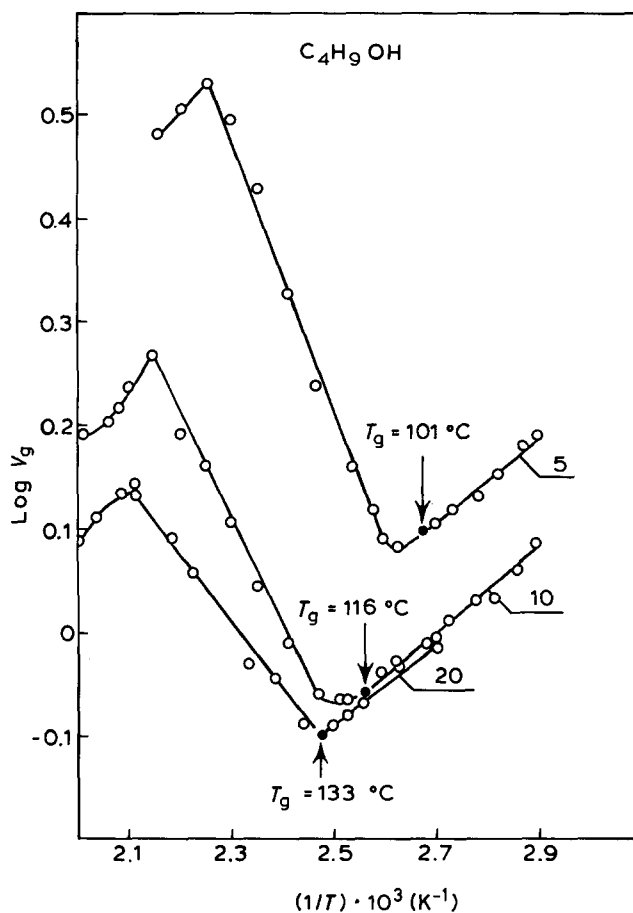
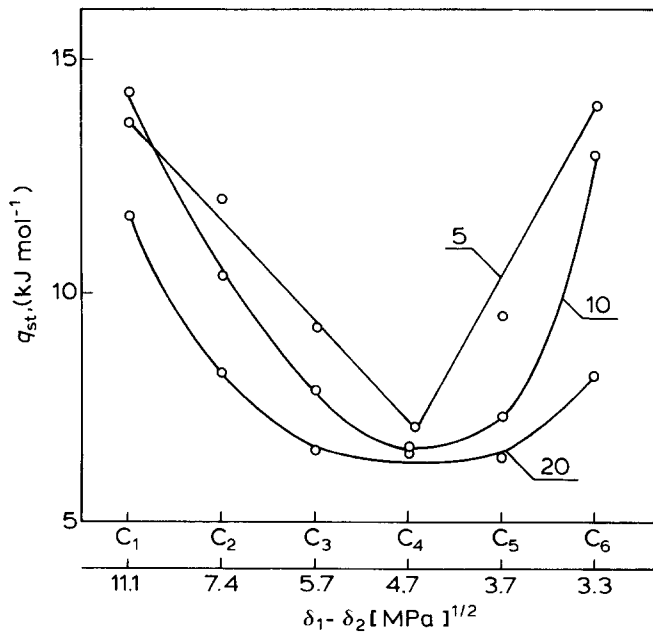


Figure 4 Retention diagram for S-DVB copolymers and butanol as molecular probe

Table 4 Regression analysis of the adsorption region of retention diagrams

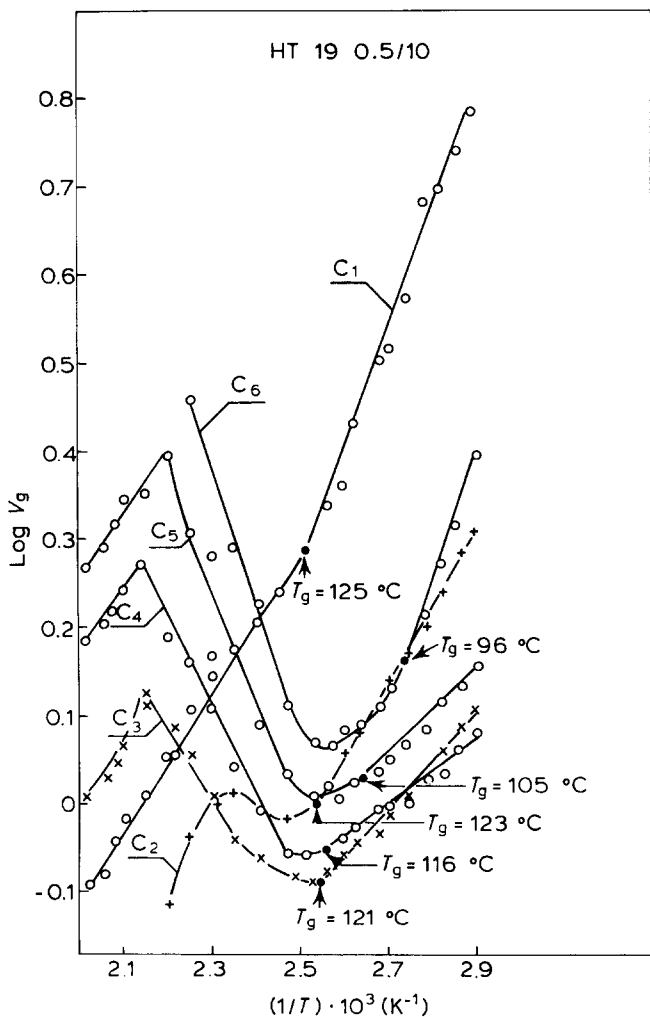
HT 19 0.5/20						
	$C_1$	$C_2$	$C_3$	$C_4$	$C_5$	$C_6$
Regression line	$1001.5 \times -2.32$	$647.1 \times -1.68$	$380.2 \times -1.05$	$369.1 \times -0.99$	$496.1 \times -1.00$	$629.7 \times -1.62$
Correlation coefficient	0.9988	0.9975	0.9999	0.9987	0.9994	0.9974
Standard deviation	0.0040	0.0046	0.0009	0.0020	0.0013	0.0043
Number of points	8	6	7	7	6	5
Confidence interval						
0.95 ±	0.010	0.011	0.002	0.005	0.004	0.011
0.99 ±	0.014	0.016	0.003	0.007	0.005	0.017
HT 19 0.5/10						
Regression line	$1360.8 \times -3.15$	$836.7 \times -2.12$	$518.8 \times -1.52$	$402.6 \times -0.91$	$540.0 \times -1.41$	$148.4 \times -3.90$
Correlation coefficient	0.9996	0.9990	0.9942	0.9975	0.9967	0.9949
Standard deviation	0.0047	0.0044	0.0069	0.0027	0.0033	0.0091
Number of points	11	10	11	8	7	5
Confidence interval						
0.95 ±	0.010	0.010	0.015	0.006	0.008	0.024
0.99 ±	0.015	0.014	0.022	0.009	0.012	0.037
HT 19 0.5/5						
Regression line	$1219.6 \times -2.73$	$1011.1 \times -2.50$	$806.7 \times -2.12$	$471.0 \times -1.17$	$881.4 \times 2.28$	$1081 \times -5.56$
Correlation coefficient	0.9994	0.9960	0.9985	0.9942	0.9981	0.9980
Standard deviation	0.0043	0.0080	0.0035	0.0036	0.0029	0.0052
Number of points	9	7	7	6	4	3
Confidence interval						
0.95 ±	0.010	0.019	0.008	0.009	0.008	0.016
0.99 ±	0.014	0.028	0.012	0.013	0.014	0.030



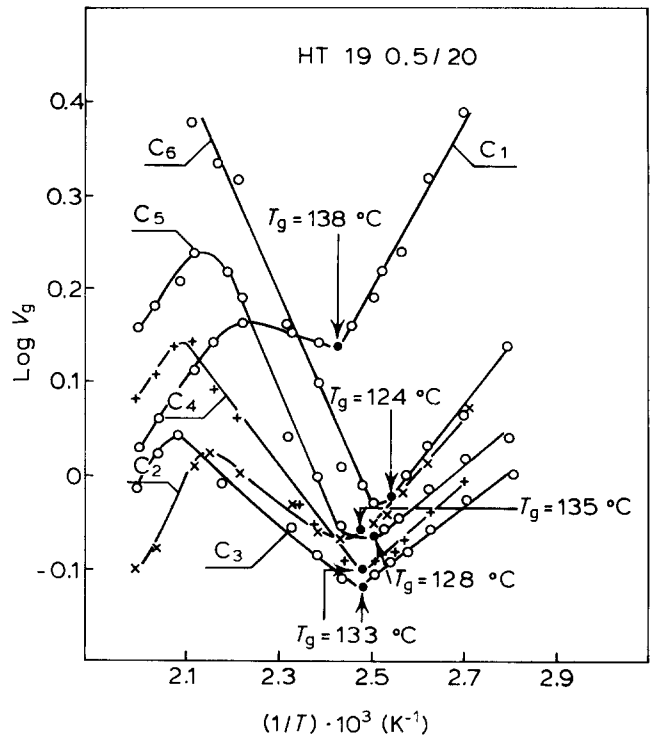
**Figure 5** Isosteric heat of adsorption,  $q_{st}$ , as a function of the chain length of alcohols used as molecular probes for S-DVB copolymers crosslinked with 5, 10, 20 wt% of DVB

The decrease of the isosteric heat of adsorption with increasing length of the alcohol chains is a striking result since for Poropak O<sup>20</sup>, as well as for graphitized carbon black<sup>29</sup>, a linear relationship between  $q_{st}$  and the number of carbon atoms in alcohols has been found. It is probable that the heterogeneity of the relatively loosely crosslinked gel allows the alcohols of smaller molecular size to penetrate and interact with a larger proportion of copolymer than is the case for alcohols with larger sized molecules. For C<sub>4</sub> the situation returns to normal, i.e. the heat of adsorption increases with increasing size of the alcohol molecules.

A deviation of a point from the straight line beyond the confidence interval of 0.95 determines an apparent  $T_g$  of copolymers (Figure 2). As has been found for other polymers, the  $T_g$  is not determined by the minimum of the retention volume in the retention diagram, but by a significant segmental mobility occurring at somewhat lower temperatures<sup>1,2</sup>. The values of  $T_g$  determined by d.s.c.<sup>33</sup> are given in Table 5. For HT 19 0.5/5 copolymer, for example, it is 103°C. The same value was obtained by using propanol as the molecular probe and it was also found that the retention volumes were the lowest for



**Figure 6** Retention diagrams for normal alcohols as molecular probes with HT 19 0.5/10 copolymer



**Figure 7** Retention diagrams for normal alcohols as molecular probes with HT 19 0.5/20 copolymer

**Table 5** Apparent values of  $T_g$

Alcohols	$T_g$ (°C) HT 19 0.5/		
	5	10	00
C <sub>1</sub>	117	125	138
C <sub>2</sub>	110	123	135
C <sub>3</sub>	102	121	133
C <sub>4</sub>	101	116	133
C <sub>5</sub>	92	105	128
C <sub>6</sub>	85	96	124
d.s.c. <sup>33</sup>	103	116	128.5

propanol. The higher alcohols act as plasticizers and reduce the  $T_g$  value. As Deshpande pointed out, not all adsorbates used in the i.g.c. method measure the true  $T_g$  value of a polymer<sup>9</sup>. Good solvents tend to reduce  $T_g$  and some of the bad solvents shift  $T_g$  towards higher temperatures.

The retention diagram for HT 19 0.5/10 copolymer is similar to that of a less crosslinked copolymer (Figure 6). For all alcohols, however, the retention volumes are lower, as also are the slopes of lines in the adsorption region.

The heats of adsorption for the copolymer crosslinked with 10% DVB are lower than those for the 5% DVB copolymers (Table 3, Figure 5). The lowest value has been found for butanol, although propanol again has the lowest retention volumes. The  $T_g$  value similar to that measured by d.s.c. determines butanol as being the probe giving the lowest value (Table 5).

The retention diagrams for HT 19 0.5/20 copolymer are presented in Figure 7. Here the retention volumes are even lower than for the less crosslinked copolymers and so are the slopes in the adsorption region that determine the heats of adsorption. It is probable that the higher surface area than that found in the less crosslinked copolymers, possibly further increased by a local microsineresis, primarily affects the retention volumes.

Again, the minimum retention volumes were measured for propanol, but the  $T_g$  corresponding to that measured by d.s.c. was determined using n-pentanol as a molecular probe.

The apparent  $T_g$  values were read out from retention diagrams as the significant deviation from the straight lines through 7 points provided the correlation coefficients were as high as 0.9999 (propanol) and 0.9987 (butanol) (Table 4). For HT 19 0.5/20 copolymer, as opposed to the other two, the apparent  $T_g$  values measured by using propanol and butanol lie at the minima of retention times.

## CONCLUSION

The i.g.c. method with n-alcohols as molecular probes can be used to estimate the temperature of first segmental mobility in S-DVB copolymers. With propanol or butanol the measured  $T_g$  values are close to those determined by d.s.c.

The relationship between the isosteric heat of adsorption and the number of carbon atoms in the alcohols is non-linear with a minimum corresponding to butanol. For all copolymers a decrease in retention time with increasing molecular weight of alcohol from  $C_1$  to  $C_3$  followed by its increase for  $C_4$  to  $C_6$  alcohols is observed.

Since such behaviour differs from that for the non-specific absorbents such as Poropak Q or graphitized carbon black, one could expect that the supermolecular structure of S-DVB copolymers has an effect on their interactions with alcohols.

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